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Rapid, High-Temperature Microwave Soldering toward A High-Performance Cathode/Electrolyte Interface

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Abstract

Solid-state lithium batteries using inorganic electrolytes are expected to revolutionize energy storage systems due to their better safety and high energy density. However, their application is greatly hindered by the poor solid-solid interface between the solid-state electrolyte (SSE) and electrodes, particularly the cathode. Herein, we report a facile strategy to address the high cathode/SSE interfacial resistance through rapid, high-temperature microwave soldering. As a proof-of-concept demonstration, we soldered a garnet-type Li$_7$La$_3$Zr$_2$O$_{12}$ (LLZO) SSE with a V$_2$O$_5$ cathode, which feature high thermal stability and suitable melting temperatures. Our microwave soldering technique can selectively melt the surface of the granular V$_2$O$_5$ and rapidly form an intact and continuous cathode layer with tightly embedded carbon black nanoparticles, leading to a remarkable 690-time increase of the electronic conductivity of cathode. Additionally, the melted V$_2$O$_5$ cathode is conformally soldered to the garnet electrolyte, resulting in a 28-fold decrease of the cathode/garnet interfacial resistance (from 14.4 kΩ·cm$^2$ to 0.5 kΩ·cm$^2$). As a result, this all-solid-state full cell displays a low overall resistance of 0.3 kΩ·cm$^2$ at 100 °C, which enables stable cyclability of the battery without the addition of liquid/polymer electrolyte. The fast microwave soldering strategy constitutes a significant step towards the development of the all-solid-state batteries.
Introduction

Solid-state lithium batteries (SSLBs) are considered a promising energy storage system due to their high energy density, longer cycle life, and improved safety compared to traditional lithium ion batteries. Solid-state electrolytes (SSEs) have enabled significant advances in SSLBs, with numerous materials investigated, including LIPON, LISCON, perovskites, antiperovskites, hydrides and garnet. These studies have resulted in an improved bulk Li-ionic conductivity of 10 mS/cm, which rivals that of standard organic liquid electrolytes. Nevertheless, large-scale commercialization of SSLBs has remained challenging due to the poor solid-solid interfacial contact between the SSE and electrodes (both anode and cathode), which creates large interfacial impedance that can be more than two orders of magnitude higher than batteries with liquid or polymer electrolytes. Furthermore, this poor and non-conformal interfacial contact can lead to high local current density, which may cause uneven Li plating/stripping and that could ultimately short circuit the battery.

Fortunately, the high electronic and Li-ion conductivity and low softening temperature (~180 °C) of the lithium metal anode provide us the opportunity to minimize the anode/SSE interfacial impedance through surface modification and interfacial engineering. For instance, by introducing a thin interphase layer between the Li anode and SSEs, the wetting interaction between them is dramatically improved, resulting in a significant decrease of interfacial resistance. However, the cathode interface has remained an issue hindering the development of SSLBs due to the low ionic conductivity of cathode materials and the poor contact between rigid granular cathode powders and stiff SSEs.
To date, limited approaches have been proposed to address the problem of the cathode/SSE interface, among which most involve the utilization of a new layer of Li-ion conductor. For example, a polymer Li-ion conductor was sandwiched between the cathode and SSE to cushion the poor contact at their interface. The crosslinked polymer with good interfacial compatibility serves as both binder and electrolyte, which enables stable cycling of the solid-state batteries. However, the soft organic polymer may introduce potential safety concerns, as well as sacrifice the device’s thermal stability and lower its energy density due to the addition of inactive materials. In response to these limitations, inorganic Li-ion conductive interlayers (e.g., Li$_4$Ti$_5$O$_{12}$, LiBO$_3$, Li$_{2.3}$C$_{0.7}$B$_{0.3}$O$_3$, and Li$_{1.4}$Al$_{0.4}$Ti$_{1.6}$(PO$_4$)$_3$) have recently been employed to improve the cathode interface, for rechargeable solid-state batteries. These ceramic interlayers, which feature good thermal and chemical stability as well as high mechanical strength, can significantly improve the battery’s safety. By sintering them at the interface between the cathode and SSE, sufficient interfacial contact is also achieved, which leads to reduced interfacial impedance and improved electrochemical performance. However, the long sintering time (usually longer than 30 min) at high temperature not only lowers the processing efficiency but also causes side reactions that compromise the battery performance. Therefore, a facile and efficient technique is needed to improve to address the cathode interface issue in order to achieve high performance SSLBs.

In this work, we demonstrate a rapid high-temperature microwave soldering strategy that enables close binding between the cathode and SSE, leading to an improved interface with significantly decreased interfacial resistance. As a proof-of-concept demonstration, we subjected the solid-
solid interface between a V$_2$O$_5$ cathode and garnet-type LLZO SSE to this microwave soldering process, due to their high thermal stability and suitable melting temperatures. The rapid microwave heating enables the selective melting and re-solidifying of the surface of V$_2$O$_5$ within a few seconds, which results in the soldering between the cathode and SSE as well as the particles in the cathode composite. As a result, the interfacial resistance between the cathode and solid electrolyte as well as the electronic resistance of cathode composite dramatically decrease. Based on the improved cathode interface, this Li/Garnet/V$_2$O$_5$ all solid-state battery displayed a low overall resistance and was successfully cycled over 60 cycles at 100 $^\circ$C, without the introduction of other interlayer or polymer binders. This work provides a facile and effective strategy to achieve a sufficient interface between the cathode and SSE, constituting a significant step towards the development of safe and high-energy-density all-solid-state batteries.

Figure 1 shows the schematic of the rapid microwave soldering process and its effect on the cathode/SSE interface. Carbon black (CB) was mixed with V$_2$O$_5$ to improve the electronic conductivity of the cathode composite. This composite was then coated onto both sides of the garnet pellet and dried to form a symmetric cell. Due to the high stiffness of the V$_2$O$_5$ grains (Figure S1), the cathode has poor physical contact with the garnet SSE as well as carbon black (Figure 1a). We then buried this cell in a carbon-black bath, which provides a high-temperature environment when subjected to microwave irradiation (Figure 1b). Rapid soldering was enabled by two-steps in this microwave heating process, including slow preheating (500 W, 9 s) and rapid heating (1000 W, 7 s). The slow preheating can prevent the garnet pellet from cracking due to the intense temperature ramping achieved from rapid heating. During the rapid heating, the temperature of the carbon-black bath increased up to ~1200 K, which is much higher than the
melting temperature of V$_2$O$_5$ (~960 K). As a result, the V$_2$O$_5$ melted from surface and penetrated into the void spaces between the solid content, resulting in solid soldering between the cathode and SSE as well as the adjacent cathode particles (Figure 1c). The short soldering time can also inhibit potential side reactions that impair the SSE and cathode materials, which is essential for achieving a high-performance solid-state battery.
Figure 1. (a) Before microwave soldering, the rigid V\textsubscript{2}O\textsubscript{5} particles are isolated with poor point-to-point contact with both the carbon black nanoparticles and SSE pellet. (b) Schematic illustration of the rapid microwave soldering process of the cathode/SSE interface. (c) The high-temperature soldering causes the V\textsubscript{2}O\textsubscript{5} to melt and penetrate into void spaces, forming good interfacial contact between the carbon black and SSE.

Figure 2a shows a photo image of the microwave soldering process in which a strong light emission can be observed due to the high temperature achieved. The emission is above the black level and within the error threshold that can be captured to calculate temperature based on the black-body theory.\textsuperscript{32} Therefore, we used a high-speed camera to record the sample’s color
images during the soldering treatment, which were then computer processed to provide the raw color of each pixel, enabling spatial- and time-resolved ratio pyrometry to obtain the temperature. The peak temperature of the soldering process reached up to ~1200 K, which lasted for 3 s until the microwave was stopped, with a small temperature fluctuation of less than ±30 K (Figure 2b). The subsequent cooling process under blowing air takes no more than 10 s, which makes the whole treatment less than 20 s long. As reported in our previous study, the temperature of microwave soldering can be controlled by adjusting the microwaving power or the amount of carbon black in the heating bath. Note that the short soldering period and stable soldering temperature is beneficial for inhibiting side reactions.

To study the effect of the microwave soldering, we first conducted a tape-testing experiment to investigate the contact between the cathode composite and garnet SSE. Specifically, the scotch tape was pasted on the cathode side and peeled off to see whether the cathode layer was remained well. Before microwave soldering, the cathode composite could be peeled off easily from the surface of the garnet, indicating insufficient interfacial contact (Figure 2c, top). In contrast, after microwave soldering, the cathode composite was bonded tightly to the garnet surface without peeling-off, demonstrating an obviously improved cathode/garnet interface (Figure 2c, bottom).

We also used scanning electron microscopy (SEM) to study the micromorphology of the interface between the V$_2$O$_5$ cathode and garnet SSE as well as the structure of cathode composite. The cross-sectional image of the unmodified cathode/SSE interface shows that the bad contact between the cathode and SSE, leaving a large interfacial gap of ~2 μm (Figure 2d), which
indicates a poor interface between the cathode and garnet. In addition, the $V_2O_5$ cathode and conductive carbon black were granularly isolated, with a loose and porous structure (Figure 2e, S2), indicating inferior physical contact within the cathode composite. However, after microwave soldering, we observed a dense cathode layer closely adhered on the garnet SSE with no noticeable gap, demonstrating significantly enhanced interfacial contact between the cathode and SSE (Figure 2f). Additionally, the top-view image shows the $V_2O_5$ particles melted together, forming a continuous matrix with the uniformly embedded conductive carbon black (Figure 2g, S3). Figure 2h shows the cross-sectional energy-dispersive X-ray spectroscopy (EDS) mapping of the soldered cathode/garnet interface, where La, Zr (from garnet), and V (from the cathode) elements were conformal distributed in garnet and cathode sides, further suggesting good interfacial contact between the cathode and garnet SSE.
Figure 2. Characterization of the rapid microwave soldering process. (a) Photo image of the high-T soldering process. A strong light emission can be observed due to the black body radiation. Inset is the photo image of the device before high-temperature soldering. (b) The temperature profile of the microwave process, showing a stable temperature (~1200 K) was maintained for 3 s. (c) Photo images of the tape-testing experiment of the unmodified V$_2$O$_5$/SSE (top) and soldered V$_2$O$_5$/SSE (bottom). The soldering treatment results in strong contact between the V$_2$O$_5$ and SSE with no cathode detachment. (d) Cross-sectional and (e) top-view SEM images of the unmodified cathode/SSE. (f) Cross-sectional and (g) top-view SEM images of the soldered cathode/SSE. (h) EDS elemental mapping of the soldered V$_2$O$_5$/SSE cross-section.

Besides the improved interfacial contact, it is also important to ensure there is no degradation of the cathode and SSE during the microwave soldering process. The grain size distributions of unmodified and microwave-treated garnet were calculated based on SEM images (Figure S4), which show negligible change after microwave soldering. The Raman spectra of both the unmodified and microwaved cathode/SSE half-cells feature the same main characteristic peaks of the cubic garnet phase (Figure 3a), demonstrating no phase change in the SSE during high-temperature soldering. Furthermore, the peak at ~160 cm$^{-1}$ in the unmodified garnet/cathode, corresponding to common Li$_2$CO$_3$ contaminant in garnet SSE, is absent in the microwaved garnet/cathode half-cell, indicating the removal of Li$_2$CO$_3$ after the microwave soldering process (Figure 3a). X-ray diffraction (XRD) patterns also confirm that the characteristic peaks of both the garnet (PDF#80-0457) and V$_2$O$_5$ (PDF#41-1426) are well preserved. The broaden of the garnet peaks are likely due to the lattice strain during the high temperature treatment$^{34}$, which
causes no degradation of Li\textsuperscript{+} conductivity as demonstrated later. Moreover, the two obvious peaks of Li\textsubscript{2}CO\textsubscript{3} (PDF#22-1141) are not present after microwave soldering (Figure 3b). The cleaning effect is also indicated by the morphology change of the garnet before and after microwave soldering (Fig. S4). The surface of the unmodified garnet was rough and covered with a layer of contamination (Figure. S4a, b), typically Li\textsubscript{2}CO\textsubscript{3},\textsuperscript{27,35,36} which disappeared after microwave soldering, revealing a clean and smooth garnet surface (Figure S4c, d). The contact angle between Li anode and Li\textsubscript{2}CO\textsubscript{3}, simulated by density functional theory (DFT), is as high as 142°, meaning a bad wettability between them. In addition, the Li\textsubscript{2}CO\textsubscript{3} contamination features a low Li\textsuperscript{+} conductivity (10\textsuperscript{−7} S/cm at 25 °C) and a low decomposition voltage (3.2 V). In this way, the removal of Li\textsubscript{2}CO\textsubscript{3} is beneficial for realizing a higher ionic conductivity and good stability for garnet SSE as well as better wettability with the lithium metal anode.\textsuperscript{37-39}

We measured the I-V curves of the cathode layer using the four-probe method to study the effect of soldering treatment on the electronic conductivity of cathode composite. Based on the I-V curves (Figure 3c), the unmodified cathode exhibits a conductivity of 0.0016 S/cm, which dramatically increases to 1.1 S/cm after soldering treatment, corresponding to a 690-fold improvement. We attribute the improved conductivity of the cathode composite to the continuous V\textsubscript{2}O\textsubscript{5} and tightly embedded nanograined carbon black (Figure S3). The high electronic conductivity benefits the charge transfer of the electrode, which can lead to better electrochemical performance.\textsuperscript{40-42} Additionally, we used electrochemical impedance spectroscopy (EIS) to study the resistance of the garnet itself and cathode/garnet interface. The bulk resistance of the unmodified garnet was first evaluated with a lithium electrode, following our previous method,\textsuperscript{43} resulting in values of 160, 62, 23, and 13 Ω·cm\textsuperscript{2} at 25, 50, 75, and 100 °C,
respectively, corresponding to ionic conductivities of $2.9 \times 10^{-4}$, $7.4 \times 10^{-4}$, $2.0 \times 10^{-3}$, and $3.5 \times 10^{-3}$ S/cm (Figure S5). As shown in Figure 3d, the overall resistance from the electrolyte and interfaces before soldering was $29.0 \ \text{k}\Omega \cdot \text{cm}^2$ at 25 °C, therefore we calculated the interface resistance to be as high as $14.4 \ \text{k}\Omega \cdot \text{cm}^2$ (interfacial resistance = (overall resistance-bulk resistance)/2). After microwave soldering, the interfacial resistance dramatically decreased to 0.5 kΩ·cm², corresponding to a more than 28-times decrease. Note that the ionic conductivity of the garnet itself showed only a slight increase of less than 0.2-fold after an identical microwave treatment, as evidenced by the EIS results (Figure S6). Therefore, we can attribute the decrease of the all-solid-state cathode/electrolyte interfacial resistance to the improved contact after microwave soldering. In addition, the diffusion impedance decreases significantly after microwave soldering, which is likely due to the conformal interconnection within the cathode layer (Figure S3).

Due to the excellent thermal stability of both the electrode and electrolyte, one promising application of this solid-state battery lies in high-temperature energy storage systems, such as thermal battery used in geothermal boreholes. Therefore, we also conducted EIS of the microwave-soldered symmetric cell at increased temperatures of 50, 75, 100 °C, which show distinct differences from that of 25 °C. Based on these spectra (Figure 3e), we calculated the cathode/electrolyte interfacial resistance to be 115, 66, and 26 Ω·cm² at 50, 75, and 100 °C, respectively (Figure 3f). Therefore, the high temperature can further facilitate the interfacial charge transfer, thus leading to fast interfacial reaction kinetics in the solid-state battery. Note that the microwave soldering technique is typically a melting-solidification process under high temperature which is also a common transformation process in other material. Therefore, it
shows great potential in welding other cathode materials with garnet SSE, especially those who exhibit good thermal stability (e.g. LiFePO$_4$).

Figure 3. Characterization of the cathode/SSE/cathode symmetric cells, using garnet SSE and V$_2$O$_5$/carbon black composite electrodes. (a) Raman spectroscopy and (b) XRD patterns of the unmodified and microwave (MW) soldered symmetric cells. (c) I-V curves of the cathode layers
and (d) an EIS plot of the unmodified and microwave-soldered symmetric cells. (e) The EIS and (f) calculated interfacial resistance of the symmetric cell after microwave soldering at temperatures of 25, 50, 75, and 100 °C.

To demonstrate the potential applications of this improved interface, we assembled an all solid-state battery by melting the lithium anode onto the electrolyte side of the microwave-soldered cathode/garnet half-cell (Figure 4a). Figure 4b shows the EIS plots of the resulting battery, in which the overall resistance displays a remarkable drop as the temperature increases. When elevating the temperature from 25 to 100 °C, the resistance from the electrolyte and interfaces decreased by 9-fold, from ~490 Ω·cm² to ~54 Ω·cm². Meanwhile, compared to the unmodified cathode/garnet cell, the overall resistance of the battery after soldering significantly decreased at all testing temperatures (Figure 4c, S7). Since the garnet pellets, electrodes and the anode/garnet interfaces in the two cases are almost identical, the dramatically decreased overall resistances can be attributed to the improved cathode/garnet interface.

Due to this improved interface, the all solid-state battery can cycle effectively between ~1–4.2 V at 100 °C. Figure 4d shows the initial charge/discharge curves of the batteries with the unmodified and microwave-soldered cathode/garnet interface at a current density of 116 mA/g. The battery with the soldered interface exhibits a significantly prolonged charge/discharge time (~15-fold, Figure 4e), which means a much higher specific capacity due to the improved cathode/garnet property. Additionally, the magnified voltage profiles showed distinct plateaus with a small overpotential for the soldered cell when compared to the battery with the unmodified cathode/garnet interface, indicating a remarkably reduced electrochemical polarization. (Figure S8).
To further study the cycling performance and rate performance, the soldered solid-state battery was cycled at 100 °C under various current densities. When cycled at 116 mA/g, the discharge capacity of the first cycle was ~114 mAh/g, which stabilized to ~80 mAh/g in the following 20 cycles (Figure 4f). Even at 174, 232, and 290 mA/g, the discharge capacity can reach 70, 60, and 48 mAh/g, respectively. When recovering the current density back to 116 mA/g, the discharge capacity returns to ~78 mAh/g, demonstrating the superior stability of the soldered cathode/garnet interface. The coulombic efficiency exceeded 90% at all the current densities suggesting a good reversibility of the soldered battery. In addition, after cycling, the EIS plots of the battery show negligible increase in overall resistance (Figure 4g). These results indicate that the solid-state battery Li/SSE/V$_2$O$_5$ featuring the microwave-soldered cathode interface exhibits superior electrochemical stability.
Figure 4. Characterization of the all-solid-state full cells. The cathode/garnet interface was microwave-soldered before assembling. (a) Schematic illustration of the all-solid-state full cell, using garnet electrolyte, the $\text{V}_2\text{O}_5$/carbon black composite cathode, and a lithium metal anode. (b) EIS and (c) comparison of overall resistance of the Li/SSE/$\text{V}_2\text{O}_5$ cells (with unmodified or soldered cathode interfaces) tested at temperatures of 25, 50, 75, and 100 $^\circ\text{C}$. (d)
Discharge/charge curves for the batteries featuring the unmodified and soldered cathode interfaces. (e) Magnified view of a certain region of (d). (f) The discharge capacity and coulombic efficiency of the Li/SSE/V$_2$O$_5$ battery with the microwave-soldered cathode interface, cycled at 100 °C. (g) EIS of the microwave-soldered cell, before and after cycling.

**Conclusion**

In summary, we reported a rapid microwave soldering strategy for overcoming the large solid-solid interfacial resistance between the cathode and ceramic electrolyte. The high-temperature microwave heating enables the fast melting/solidifying of the cathode to produce good interfacial contact with SSE, while the short soldering duration inhibits detrimental side reactions. The high-temperature soldering process can also boost the electronic conductivity of the cathode composites, which is important to improve the performance of the battery. As a demonstration, we used this strategy to achieve conformal contact between a V$_2$O$_5$ cathode and garnet solid-state electrolyte, as well as close interconnection between the particles in the cathode composite, which lead to significantly reduced interfacial impedance and cathode resistance. As a result of these advantages, the all-solid-state Li/Garnet/V$_2$O$_5$ battery based on the microwave-soldered cathode interface displayed an overall resistance as low as 0.3 kΩ·cm$^2$ at 100 °C and exhibited a superior electrochemical performance, with a discharge capacity of 80 mAh/g when cycled at 116 mA/g for 20 cycles and 48 mAh/g when cycled at 290 mA/g for 10 cycles. We note this microwave soldering method is not limited to V$_2$O$_5$ and garnet but can be extended to other cathode and electrolyte systems with suitable melting temperature, thus providing a promising strategy for the practical application of an all-solid-state battery.
Experiments

Assembly of the cells.
The garnet SSE was synthesized by following a previous method without Al doping.\textsuperscript{37} The cathode/garnet/cathode symmetric cell was fabricated by a simple dip coating method. Typically, a mixture of V\textsubscript{2}O\textsubscript{5} powders (99.6\%, Sigma-Aldrich) and carbon black with mass ratio of 86:14 were ground in a mortar for 30 min before being dispersed into N-methyl-2-pyrrolidone (>99.7\%, Sigma-Aldrich) to form a 20 mg/mL cathode dispersion. Then, the cathode dispersion was dip-coated onto both side of a freshly polished garnet pellet, followed by drying in a 60 °C oven for 5 h. The mass loading of the cathode composite was controlled to be ~2 mg/cm\textsuperscript{2}, ~5 μm. The Li/garnet/Li symmetric cell was fabricated in an Ar-filled glovebox by smearing molten Li onto both sides of the freshly polished garnet, during which a small amount of Sn was mixed with Li (the mass ratio of Sn to Li was ~1/4) to minimize the Li/garnet interfacial impedance, following a previous work.\textsuperscript{38} The all solid-state battery was assembled by similarly smearing the molten lithium on to the garnet side of the cathode/garnet half-cell.

Rapid microwave soldering.
The as-prepared cells were buried into a bath of carbon black (Ketjen black EC-600JD,160 mg) in a vial, which was then directly placed in a domestic microwave oven (Panasonic, NN-H965BFX). Here, carbon black serves as a heating source to provide a high-temperature environment for the rapid soldering when subjected to a two-step microwave radiation process. Specifically, 500 W microwave radiation was first applied for 9 s to preheat the sample, during which the temperature was <1000 K, as no light emission was observed in this process.\textsuperscript{47} After that, 7 s of 1000 W microwave radiation was immediately applied to generate high temperature
(~1200 K) for the rapid soldering. The temperature of the soldering process was evaluated, following a previous method\textsuperscript{32, 33} by fitting the spectrum of the light emission with black-body radiation equation, which was captured by a high-speed camera (Vision Research Phantom Miro M110). As technique relies on the visible light emission (typically >1000 K) from black-body radiation, we measured the peak temperature of the soldering process, which occurred in the last 3 s of microwave radiation.

**Materials characterization**

The morphologies and elemental mapping were examined on a field-emission SEM (Hitachi SU-70). The XRD patterns were obtained on a D8 Advance system (Bruker AXS, WI) using Cu K\(\alpha\) radiation. Raman characterization was conducted on a Horiba Jobin Yvon using a 532 nm solid-state laser with a 4 s integration time for 4 repeated measurements. The I-V curve was measured with a typical four-point probe method using a Keithley 2400 as the power supply unit.

**Electrochemical characterization**

Electrochemical performance was tested on a Bio-Logic potentiostat. The EIS plots were tested with 20 mV AC amplitude in the frequency from 0.1 Hz to 1 MHz. The cycling measurement was conducted at current densities from 116 mA/g to 290 mA/g, with a voltage of ~1.0–4.2 V. All the electrochemical tests were carried out in an Ar-filled glovebox with a box furnace to control the testing temperature.
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Author contributions

L. Hu, G. Zhong, and C. Wang designed the experiments. G. Zhong conducted the microwave soldering experiments. C. Wang, R. Wang, and W. Ping synthesized the garnet electrolyte. G. Zhong, R. Wang, and W. Ping carried out the electrochemical characterization and analysis. H. Qiao, M. Cui, and Y. Zhou performed the material characterization. X. Wang, D. J. Kline, and M. R. Zachariah contributed to the temperature measurement. G. Zhong, C. Wang, and L. Hu collectively analyzed the data and wrote the paper.
Supporting Information for
Rapid, High-Temperature Microwave Soldering toward a High-Performance Cathode/SSE Interface

Figure S1. SEM image of V$_2$O$_5$. The high stiffness of the V$_2$O$_5$ grains at room temperature is adverse to forming good contact with the garnet SSE or carbon black.
**Figure S2.** SEM image of unmodified cathode composite layer, in which the cathode V$_2$O$_5$ and conductive carbon black are granularly isolated with a loose and porous structure.
Figure S3. SEM image of the soldered cathode composite layer, in which the V$_2$O$_5$ particles are melted together, forming a continuous matrix, with the conductive carbon black uniformly embedded within.
**Figure S4.** The cross-section SEM images of the (a, b) unmodified and (c, d) soldered garnet. The cross-section of the unmodified garnet is rough and covered with a layer of contamination, which disappeared after microwave soldering, revealing a clean and smooth garnet surface.
Figure S5. a. EIS of the Li/Garnet/Li cell at temperatures ranging from 25–100 °C. b. Arrhenius plot of the garnet ionic conductivity.
Figure S6. EIS of the Li/Garnet/Li symmetric cell using the unmodified garnet and microwave (MW)-treated garnet, which show only a slight increase in resistance.
Figure S7. EIS of the all solid-state battery based on the unmodified cathode/garnet interface.
**Figure S8.** The voltage profiles of the first cycle of the all solid-state batteries. The battery based on the soldered cathode/garnet interface shows distinct plateaus with small overpotential, while the battery with the unmodified cathode/garnet interface shows no obvious plateau with large overpotential.
Declaration of interests

☒ The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

☒ The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

No potential competing interests.